

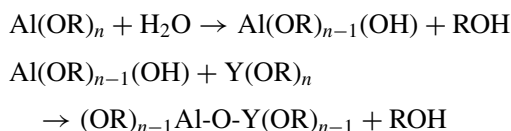
# Preformed sol-gel synthesis and characterization of $\text{YAIO}_3$

P. A. TANNER\*, PO-TAK LAW, KA-LEUNG WONG, LIANSHE FU  
 Department of Biology and Chemistry, City University of Hong Kong, Tat Chee Avenue,  
 Kowloon, Hong Kong S.A.R., People's Republic of China  
 E-mail: bhtan@cityu.edu.hk

The synthesis of  $\text{YAIO}_3$  is described from commercially-available yttria and alumina preformed sols which have nanometer particle-size ranges. The products have been characterized by X-ray diffraction (XRD), diffuse reflectance infrared Fourier transform (DRIFT) spectra, thermogravimetry—derivative thermogravimetry—differential thermal analysis (TG-DTG-DTA), scanning electron microscopy (SEM) as well as CHN-analysis, and are similar to those prepared by the alkoxide, nitrate and citrate routes. The proposed method is a simple, cheap and rapid synthetic route. © 2003 Kluwer Academic Publishers

## 1. Introduction

The conventional preparation of  $\text{YAIO}_3$  (YAP) involves the high-temperature (ca. 2000°C) Czochralski growth in an inert atmosphere [1]. Other than the hydrothermal and glycothermal methods [2], sol-gel syntheses have employed the conventional alkoxide route [3, 4], as well as acetate [5], citrate [6], and some other methods [7–9]. Although high-purity alkoxides are available, the hydrolysis-condensation reactions:



do not permit the step-by-step control of sol and gel formation.

The motivation of the present study was to explore the simple, low-temperature, rapid method of preparation of YAP. This is the first, crucial stage of a project to dope the resulting powders with lanthanide ions and to examine the luminescence of the phosphors under laser excitation. This method involved the use of commercial preformed sols in different size ranges on the nanometer scale. These materials are widely available, cheaper and more easily stored and used than alkoxides. Preformed sols are stable mixtures of dispersed particles which enable the sol and gel processes to be isolated and performed in two steps, offering potential control over the particle size, crystallinity and microstructure of the product. The preformed sol route has other advantages that *different-size* precursors are readily available in aqueous solutions, employed herein, or as dispersed powders.

## 2. Experimental

Preformed sols of 10 nm yttria and 90, 94, 180 and 415 nm alumina dispersed in aqueous solutions were

donated by Nyacol Products Inc. USA, and by Condea Vista Company, USA. Some characteristics of these sols are listed in Table I. YAP powders were synthesized as follows:

### 2.1. Preformed sol method

In a typical preparation 5 cm<sup>3</sup> of 10 nm yttria was added, with vigorous stirring, to the determined volume of the alumina sol, (mol ratio Y:Al 1:1). The mixed sol was then stirred for 10 min to homogenize it. 0.5 cm<sup>3</sup> of 1 M  $\text{HNO}_3$  was added over 1 min, and a vortex usually formed in the solution under rapid stirring within 1–5 min. The white wet gel mixture was stirred for a further 10 min, left at room temperature for up to 3–4 h, then dried for 10 h at ca. 110°C, prior to grinding into powder using a ceramic pestle and mortar. The white powder was subsequently heated for 2 h at higher temperatures between 400 to 1500°C, in ceramic or Pt crucibles.

### 2.2. Preformed yttria sol-aluminium nitrate method

A weighed amount of  $\text{Al(NO}_3)_3 \cdot 9\text{H}_2\text{O}$  was dissolved in water and 5 cm<sup>3</sup> 1 M  $\text{HNO}_3$  was added. The clear mixture was stirred for 5 min. and then added dropwise to the preformed 10 nm  $\text{Y}_2\text{O}_3$  sol that was rapidly stirred in a separate beaker. Subsequent procedures were as in method 1.

### 2.3. Nitrate-citric acid method

The appropriate amount of solid  $\text{Al(NO}_3)_3 \cdot 9\text{H}_2\text{O}$  was added with stirring to  $\text{Y(NO}_3)_3$  solution in a beaker. Solid citric acid was then mixed (mol ratio citric acid: metal ion 3:1). The mixture was evaporated at 85°C on a water bath until a pale yellow transparent, viscous gel

\*Author to whom all correspondence should be addressed.

TABLE I Properties of yttria and alumina preformed sols<sup>a</sup>

Property Product name	Alumina sol					Yttria sol Nyacol Y <sub>2</sub> O <sub>3</sub>
	Dispal 11N7-12	Dispal 14N4-25	Dispal 18N4-20	Dispal 23N4-20	Nyacol Al20DW	
Al <sub>2</sub> O <sub>3</sub> or Y <sub>2</sub> O <sub>3</sub> (wt%)	12	25	20	20	20	14
Air-dried solids content (wt%)	14	30	24	24	–	–
Surface area (m <sup>2</sup> g <sup>-1</sup> )	120	140	180	230	–	–
Average dispersed particle size (nm)	415	180	94	90	50	10
Viscosity (cP)	<50	<50	<50	<50	10	10
pH range of dispersion	6.5–7.2	3.8–4.0	3.7–3.9	4.3–4.7	4.0	7.0
NO <sub>3</sub> <sup>-</sup> (wt%)	0.02	0.22	0.29	0.38	–	<sup>b</sup>

<sup>a</sup>Technical data from Vista Chemical Company, PQ Corporation and Nyacol Products Inc.

<sup>b</sup>Counter ion is 2.5 wt% acetate.

was obtained (ca. 3–4 h). The gel was dried at 130°C for 24 h. Subsequent procedures were as in method 1.

#### 2.4. Alkoxide route

The preparation of YAP by the conventional alkoxide route used aluminium isopropoxide and yttrium isopropoxide (Gelest, Inc.). The preparations were conducted in a dry box and utilized dry ethanol or propan-2-ol as solvents, following the procedure of Jung *et al.* [4].

In all of the above preparations, the doping of YAP with lanthanide ions was achieved by adding the appropriate amounts of Ln(NO<sub>3</sub>)<sub>3</sub> solutions at the first stage.

#### 2.5. Measurements

Powder X-ray diffraction (XRD) patterns were recorded in the range  $2\theta = 10\text{--}80^\circ$ , using a Philips X'pert X-ray diffraction system, with Cu K<sub>α</sub> radiation. Thermogravimetry—derivative thermogravimetry—differential thermal analysis (TG-DTG-DTA) scans

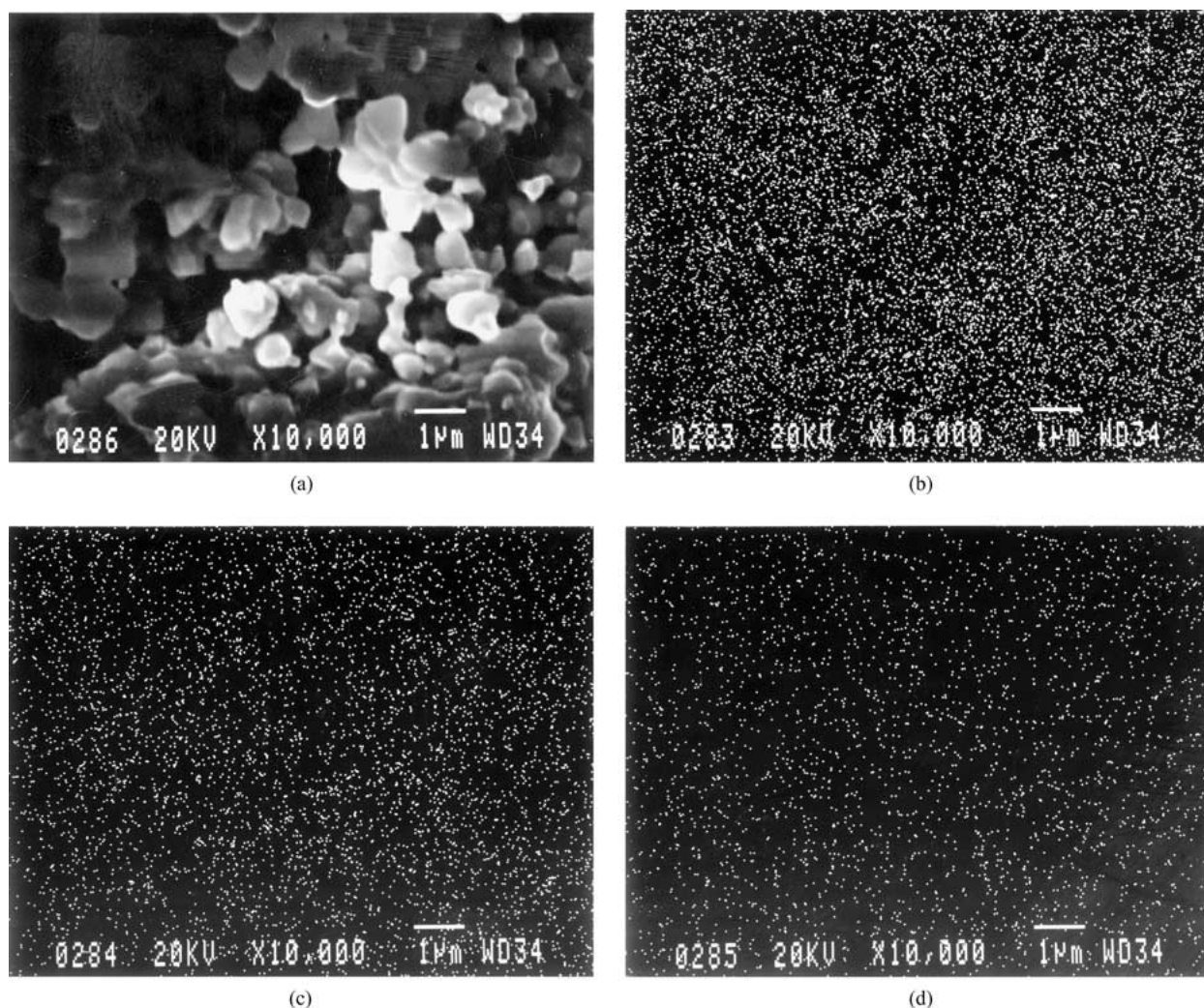


Figure 1 (a) Scanning electron micrograph of 10Y<sub>2</sub>O<sub>3</sub>-50Al<sub>2</sub>O<sub>3</sub>-(Y<sub>0.95</sub>Er<sub>0.05</sub>Al)-1.0A pre-heated at 1500°C showing the electron density mappings of (b) Y; (c) Al; and (d) Er.

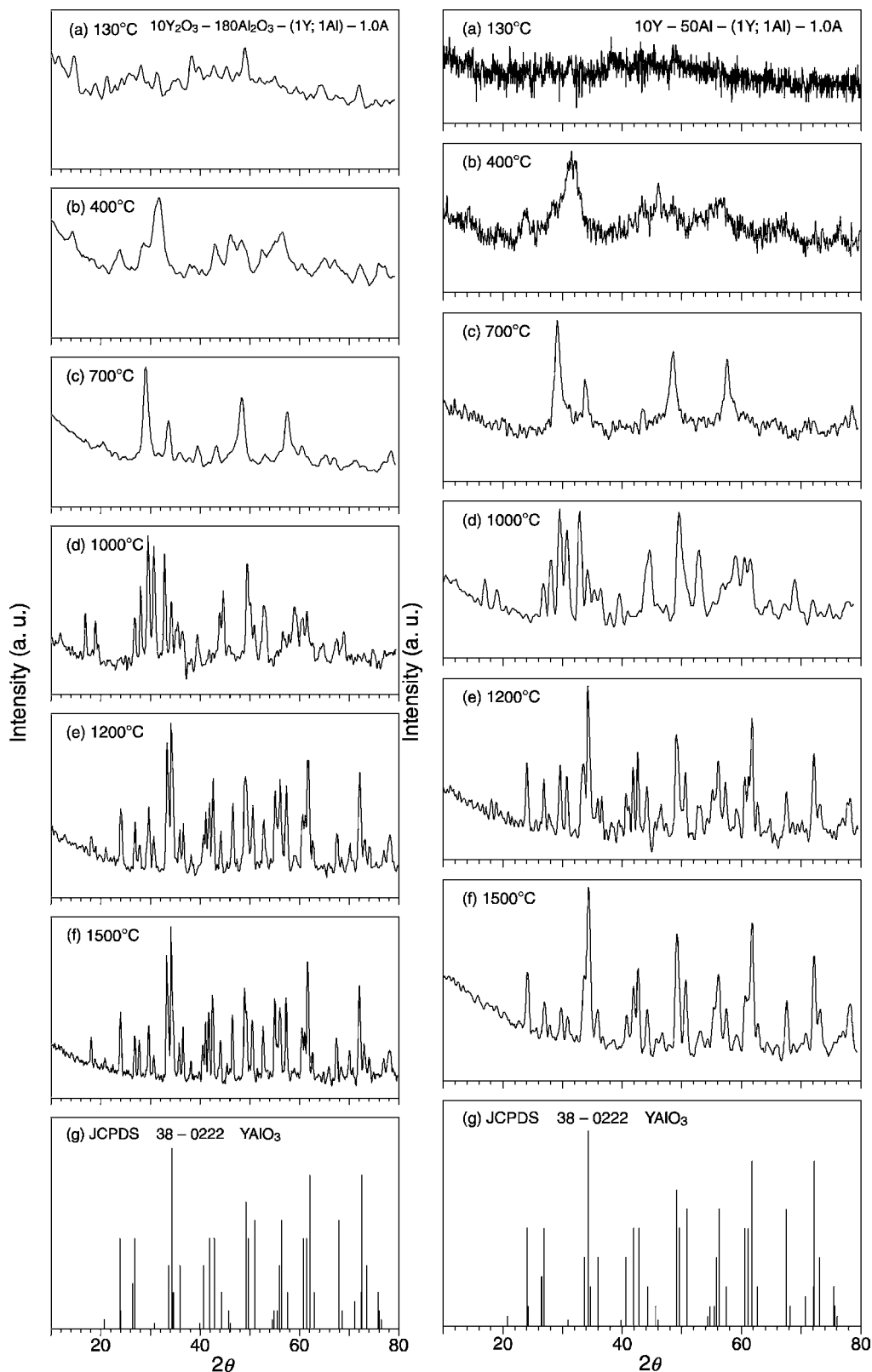


Figure 2 X-Ray diffractograms at stated temperatures of yttria-alumina products prepared from  $10\text{Y}_2\text{O}_3$  and  $50\text{Al}_2\text{O}_3$  or  $180\text{Al}_2\text{O}_3$ .

were performed between  $40\text{--}1000^\circ\text{C}$  under a He or  $\text{O}_2$  atmosphere ( $300\text{ cm}^3\text{ min}^{-1}$ ), with a heating rate  $20^\circ\text{C min}^{-1}$ , using a Seiko Instruments SSC5200H Thermal Analysis System. Fourier transform infrared (FTIR) spectra were recorded by a Perkin-Elmer (PE) Spectrum 1000 instrument, in Kulbelka-Munk (KM) Units, from samples dispersed in KBr (3% by mass), using the PE diffuse reflectance Fourier transform

infrared spectra (DRIFTS) Accessory. CHN-analyses of powders were performed upon 1–10 mg mass of samples using a Vario EL CHN Analyzer equipped with an autosampler. Ethylenediaminetetraacetic acid was used as calibration standard. Scanning electron micrographs (SEM) were obtained from gold or graphite coated samples using a Jeol 820 Instrument. Energy dispersive X-ray (EDX) analysis was used to

identify the compositions of Y, Al and Er in individual particles, or to map the concentrations over a selected area and compare with the direct SEM image.

### 3. Results and discussion

Fig. 1a shows a scanning electron micrograph of 5 mol%  $\text{Er}^{3+}$ -doped  $\text{YAlO}_3$  prepared from  $10\text{Y}_2\text{O}_3$  and  $50\text{Al}_2\text{O}_3$ , and heated to  $1500^\circ\text{C}$ . The particle sizes range from about 20 nm to  $30\ \mu\text{m}$ . No attempt was made to prepare nano-scale materials, which would involve an additional centrifugation step to separate large from small particles [10]. Fig. 1b–d show the homogeneity and different concentration regimes of Y, Al and Er from the electron density maps.

TG-DTG-DTA characterization under a He atmosphere of the sol-gel powders previously dried at room temperature showed that five distinct endothermic events take place between  $77$  to  $486^\circ\text{C}$ . Each one is associated with a mass loss, with the final mass being 56% of the initial mass. The major differences of the TG-DTG-DTA under an  $\text{O}_2$  atmosphere from that under He, are the change from endothermic to exothermic nature for the mass loss near  $400^\circ\text{C}$ , and the greater overall loss in mass. The lower-temperature mass losses are associated with loss of water, whereas those near  $400^\circ\text{C}$  are assigned to the decomposition of organic matter giving CO in He or  $\text{CO}_2$  in  $\text{O}_2$  atmosphere. The nature of the organic matter became clear from the CHN analysis of material prepared either only from the  $\text{Al}_2\text{O}_3$  sols, or from the  $\text{Y}_2\text{O}_3$  sol, alone. In the first case, the carbon content was  $<0.6\%$ , whereas it was  $13.6 \pm 0.3\%$  for the  $\text{Y}_2\text{O}_3$  powder which had been pre-heated to  $110^\circ\text{C}$ , so that the origin is attributed to the acetate present in the original  $\text{Y}_2\text{O}_3$  preformed sol. Chemical analyses of samples of yttria-alumina powders prepared by method 1, and subsequently heated to  $110^\circ\text{C}$ , showed the presence of about 11% C, which dropped to  $<2\%$  for the same samples after heating to  $400^\circ\text{C}$ ,  $<0.5\%$  for those after heating to  $800^\circ\text{C}$ , and  $<0.05\%$  for those to  $1200^\circ\text{C}$ .

Yttria-alumina samples prepared from preformed sols were investigated by XRD, and Fig. 2 shows diffractograms for two preliminary samples prepared from  $10\text{Y}_2\text{O}_3$  and  $50\text{Al}_2\text{O}_3$  or  $180\text{Al}_2\text{O}_3$ . Broad features were observed for the powders heated at  $130^\circ\text{C}$  or  $400^\circ\text{C}$ , indicating amorphous phases. The three most prominent lines at  $700^\circ\text{C}$ , are at similar locations to the three lines in the  $\text{Y}_2\text{O}_3$  sol heated at this temperature, showing that a mixture of yttria and alumina is still present. At  $700^\circ\text{C}$ , these lines are progressively sharper for increasing  $\text{Al}_2\text{O}_3$  particle sizes, showing that there is evidently a memory effect at this temperature. Also, the patterns of samples heated at  $700^\circ\text{C}$  indicate more amorphous materials from methods 2 and 3, compared with method 1, presumably representing better mixing of the components. However, at and above  $1000^\circ\text{C}$ , new lines progressively sharpen up to  $1500^\circ\text{C}$ , and the powder diagrams are identified as  $\text{YAlO}_3$ . The X-ray patterns for powders heated at  $1200^\circ\text{C}$  or above, whether prepared by methods 1, 2 or 3, were similar to the JCPDS file for YAP. These samples were prepared

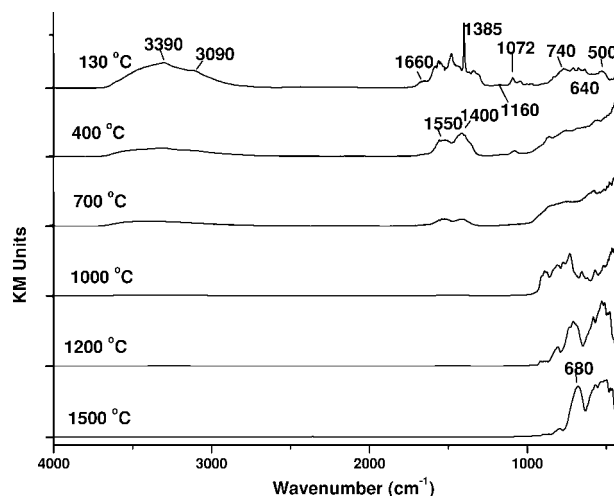


Figure 3 DRIFT spectra of gels prepared from 10 nm yttria and 50 nm alumina preformed sols (1Y:1Al) heated at various temperatures from 130 to  $1500^\circ\text{C}$ .

on a small-scale with sols  $<0.5\ \text{cm}^3$  in volume, and the error in volume measurement is fairly critical and leads to the presence impurity phases, such as YAG. Larger-scale preparations (using  $5\ \text{cm}^3$   $10\text{Y}_2\text{O}_3$  as in 2.1.) did not show other phases, as tested by the more sensitive technique of site-selective luminescence of lanthanide ion-doped samples.

DRIFT spectra of the powders prepared by method 1 from 10 nm yttria and 50 nm alumina, which were then heated at  $130^\circ\text{C}$  (Fig. 3), show bands at  $\sim 3390$ ,  $3090\ \text{cm}^{-1}$  (O–H stretching),  $1660$  (H–O–H bend),  $1160$ ,  $1072\ \text{cm}^{-1}$  (O–H bend),  $740$ ,  $640$ ,  $500$  (Al–O stretch and water librations) which are present in the spectrum of the alumina preformed sol treated alone in the same manner, to give boehmite,  $\text{AlO}\cdot\text{OH}\cdot n\text{H}_2\text{O}$ , at  $130^\circ\text{C}$ . The remaining bands are present in the spectra of the yttria sol, treated alone and heated to  $130^\circ\text{C}$ . In particular, the features at  $1550$  and  $1400\ \text{cm}^{-1}$  ( $\text{COO}^-$  antisymmetric and symmetric stretch, respectively) arise from acetate groups. The sharp peak at  $1385\ \text{cm}^{-1}$  is due to surface-adsorbed  $\text{NO}_3^-$ , and has previously been observed in samples of urban particulate matter and hydroxylated  $\text{Al}_2\text{O}_3$  exposed to  $\text{NO}_2$  [11]. The  $130^\circ\text{C}$  spectra of materials prepared using the different-size  $\text{Al}_2\text{O}_3$  precursors are similar. On heating above  $130^\circ\text{C}$ , considerable changes occur in the spectra (Fig. 3), but these again represent a mixture of the spectra of the *individual* yttria and alumina materials. However, the spectrum of YAP is evident at  $1000^\circ\text{C}$ , albeit with some water/hydroxide remaining, but this is not present at  $1200^\circ\text{C}$  or  $1500^\circ\text{C}$ . As expected, the DRIFT spectra of samples prepared by methods 2 and 3, and subsequently heated to  $130^\circ\text{C}$ , are different from Fig. 3, but the final products at  $1000$ – $1200^\circ\text{C}$  exhibit similar spectra. Although Raman data are available for  $\text{YAlO}_3$  [12] and the calculation of the zone centre phonons has been performed, the only infrared data of  $\text{MAIO}_3$  ( $M = \text{Y, Nd}$ ) are from reflection spectra [13, 14]. The highest energy transverse optic mode is at  $676\ \text{cm}^{-1}$  [13] or  $681$ – $684\ \text{cm}^{-1}$  [12], in essential agreement with the lowest spectrum in Fig. 3.

The preparation of  $YAlO_3$  is normally beset by the presence of different phases [15], particularly at high temperatures [16]. In summary, the preparation from preformed sols can give homogeneous products provided that accurate amounts of reactants are used; that these are thoroughly mixed and stirred in the gel forming process; and then subsequently ground, powdered and heated to 1200°C.

### Acknowledgement

This work was supported by the Research Grants Council of Hong Kong, CERG Grant CityU 1114/00P.

### References

1. M. J. WEBER, M. BASS, K. ANDRINGA, R. R. MONCHAMP and E. COMPERCHIO, *Appl. Phys. Lett.* **15** (1969) 342.
2. M. INOUE, H. OTSU, H. KOMINAMI and T. INUI, *J. Amer. Ceramic Soc.* **74** (1991) 1452.
3. Z. JIANG and W. E. RHINE, *Mater. Res. Soc. Symp. Proc.* **286** (1993) 21.
4. J. M. JUNG and S. B. PARK, *J. Mater. Sci. Lett.* **15** (1996) 2012.
5. G. GOWDA, *ibid.* **5** (1986) 1029.
6. M. KAKIHANA and T. OKUBO, *J. Alloys Cmps.* **266** (1998) 129.
7. Y. IDA, A. TOWATA, T. TAKAHISA and M. FURUKAWA, *Vib. Spectr.* **19** (1999) 399.
8. R. S. HAY, *J. Mater. Res.* **8** (1993) 578.
9. M. F. YAN, T. C. D. HUO and H. C. LING, *J. Electrochem. Soc.* **134** (1987) 493.
10. P. K. SHARMA, M. H. JILAVI, R. NASS and H. SCHMIDT, *J. Lumin.* **82** (1999) 187.
11. P. T. CUNNINGHAM, S. A. JOHNSON and R. T. WANG, *Environ. Sci. Technol.* **8** (1974) 131.
12. H. C. GUPTA and P. ASHDIR, *J. Solid State Chem.* **146** (1999) 287.
13. M. J. WEBER, *Phys. Rev. B* **8** (1973) 54.
14. P. ALAIN and B. PIRIOU, *Phys. Stat. Sol. (b)* **43** (1971) 669.
15. O. FABRICHNAYA, H. J. SEIFERT, T. LUDWIG, F. ALDINGER and A. NAVROTSKY, *Scand. J. Metall.* **30** (2001) 175.
16. J. L. CASLAVSKY and D. J. VIECHNICKI, *J. Mater. Sci.* **15** (1980) 1709.

Received 25 March  
and accepted 27 August 2003